Extreme ultraviolet interference lithography at the Paul Scherrer Institut

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Abstract. We review the performance and applications of an extreme ultraviolet interference lithography (EUV-IL) system built at the Swiss Light Source of the Paul Scherrer Institut (Villigen, Switzerland). The interferometer uses fully coherent radiation from an undulator source. 1-D (line/ space) and 2-D (dot/hole arrays) patterns are obtained with a transmission-diffraction-grating type of interferometer. Features with sizes in the range from one micrometer down to the 10-nm scale can be printed in a variety of resists. The highest resolution of 11-nm half-pitch line/space patterns obtained with this method represents a current record for photon based lithography. Thanks to the excellent performance of the system in terms of pattern resolution, uniformity, size of the patterned area, and the throughput, the system has been used in numerous applications. Here we demonstrate the versatility and effectiveness of this emerging nanolithography method through a review of some of the applications, namely, fabrication of metallic and magnetic nanodevice components, self-assembly of Si/Ge quantum dots, chemical patterning of self-assembled monolayers (SAM), and radiation grafting of polymers. © 2009 Society of Photo-Optical Instrumentation Engineers. [DOI: 10.1117/1.3116559]

Subject terms: nanolithography; extreme ultraviolet; interference lithography; metal nanowire; quantum dot; magnetic nanostructure; self-assembled monolayer; polymer grafting.

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1 Introduction

Extreme ultraviolet interference lithography (EUV-IL) is a newly emerging nanolithography method that combines the advantages of a parallel fabrication process with high resolution. These features make it an attractive tool for researchers who are increasingly in need of a nanopatterning capability that is beyond what is available from other methods such as photolithography, electron beam lithography (EBL), and scanning probe lithography, in terms of resolution or throughput. After early interference lithography experiments performed in the vacuum UV¹ and EUV regions,² dedicated EUV-IL tools have been set up using synchrotron^{3,4} and laser sources.⁵ Such tools using plasmabased EUV sources are also being considered.⁶

In most reported EUV-IL work, the wavelength of choice has been 13.4 nm. The reason behind this choice is manifold. First, the possible next-generation EUV lithography (EUVL) technology⁷ that is being developed for fabrication of future semiconductor devices uses this wavelength. The EUV-IL method has played an important role in the development of high-resolution resists for this technol-

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ogy by making high-resolution exposures available long before EUV projection tools become available.^{8,9} Second, it is relatively easy to fabricate diffraction gratings with sufficiently high diffraction and transmission efficiency at this wavelength on silicon nitride membranes. Finally, the short inelastic mean free path of photoelectrons created by EUV photons in a resist at this energy (92.5 eV) means that the blurring effect of photoelectrons will be insignificant down to sub-10 nm resolution. Of course, we should also mention that this wavelength allows an ultimate half-pitch (hp) resolution of below 4 nm, which can be as small as a quarter of the wavelength (λ):

$$hp = \frac{\lambda}{4\sin(\theta/2)},$$

where θ is the angle between two diffracted beams.

Several different EUV-IL schemes have been realized or proposed. A Lloyd's mirror interferometer, which is a wavefront division method, was used to expose linear gratings and 2-D patterns in a double exposure scheme.^{5,10,11} Amplitude division interferometers, one involving beamsplitter gratings with two recombining mirrors¹² and another that has two cascaded gratings,¹³ were also recently demonstrated. The EUV-IL system at the Paul Scherrer Institut (PSI) in Villigen, Switzerland,⁴ whose application is the subject of this review, uses a diffraction-grating-based wavefront division scheme.^{2,3} We describe this scheme in more detail in the following section. In addition to two and multiple beam interference methods, a holographic approach was also demonstrated.³

2 Extreme Ultraviolet Interference Lithography System at the Paul Scherrer Institut

The EUV-IL system at the XIL beamline of the Swiss Light Source (SLS) uses light that is generated by a 42-pole undulator with a 212 mm period, which is inserted into the 2.4 GeV electron storage ring. The SLS, as a third generation synchrotron, is constructed to provide a spatially coherent beam for photon energies of up to about 100 eV, thanks to the small size and divergence of the electron beam.¹⁴ The spatial coherence is essential for the basic wavefront division interference scheme that is used at this beamline. The temporal/longitudinal coherence of the light emanating from the source is related to the spectral bandwidth $\Delta\lambda/\lambda$. For an undulator, this is related to the number of magnetic poles, which translates to a rather large spectral width of about 2.5% for our source. To get a narrower bandwidth, the beam can be filtered using a monochromator. However, this leads to a significant loss of photon flux. The diffraction-grating-based interference schemes described next are achromatic. Therefore in most experiments the interferometric exposures are performed using the full bandwidth of the undulator radiation without additional spectral filtering. However, the beam is filtered by a pinhole to guarantee full spatial coherence at the interferometer. The resulting beam at the interferometer has a width of about 3 to 5 mm [full width at half maximum (FWHM)], which depends on the diameter of the pinhole used for spatial filtering. The photon flux incident on the interferometer is stable, thanks to the stability of the source



Fig. 1 The interference schemes of two- and four-grating designs to expose line and hole/dot patterns, respectively.

and the beamline optics, as well as the top-up mode of operation of the SLS storage ring. With the typical flux of 20 to 50 mW/cm², dot and line patterns in polymethyl methacrylate (PMMA) resist can be exposed within 10 sec. The beam is switched on and off by a fast mechanical shutter placed in front of the exposure chamber. Substrates ranging from several square centimeters chips up to 200 mm wafers can be exposed.

The basic interference scheme that is used at the PSI system is illustrated in Fig. 1. The spatially coherent incident beam is diffracted by linear gratings that are patterned on a single, partially transparent membrane. The resulting mutually coherent beams interfere at a certain distance along the beam direction. The interference fringes are used to expose a pattern in a resist film. The schemes in Fig. 1 create periodic patterns that have half the period of the interferometer grating. The size and shape of each exposure pattern is the same as the size and shape of the grating. Pairs of diffraction gratings with different periods can be placed on a single mask and the resulting pattern printed simultaneously in one exposure step. In this way, patterns from the micrometer down to the nanometer scale can be created in a single exposure in EUV-IL.

Two linear gratings are used to create lines, while hole/ dot patterns can be made in one exposure step by the interference of three or more diffracted beams (Fig. 1).¹⁵ More complex structures such as rings and crosses can be made by eight-beam interference.¹⁶ Depending on the design of the mask, the relation between the period of the diffraction gratings and the fringe pattern varies. For example, in a four-beam scheme, demagnification factors of 2 or $\sqrt{2}$ (in terms of pattern period) are obtained depending on the relative phases of the interfering beams.¹⁶ This demagnification factor is very helpful for the mask fabrication. For example, 200 nm period gratings written on a mask will expose 100 nm period lines on the sample. Examples of two- and four-beam masks mounted on stainless steel rings are shown in Fig. 2. The masks are prepared using electron beam lithography (EBL) and reactive ion etching pattern transfer processes on 100-nm Si₃N₄ membranes coated with a thin Cr film.¹⁷ The gratings are written in a resist with EBL and later etched into the Cr film. The thickness of the grating metal layer, the grating duty cycle, and the material (e.g., Cr) determine the EUV diffraction efficiency. Masks fabricated in the same way have been also used in other EUV-IL systems.^{3,1}



Fig. 2 (a) Two-beam and (b) four-beam interference masks prepared for exposures. The masks consist of a square silicon chip on frame with a $3 \times 3 \text{ mm}^2 \text{ Si}_3 \text{N}_4$ membrane in the center where the gratings are patterned.

Different periods and patterns can be included on a single mask. As an example, the layout of a mask that is designed to expose sets of 800 μ m-long lines with six different half-pitches in the range of 20 to 50 nm is shown in Fig. 3(a). In addition, these long lines can be cleaved in parallel to obtain cross sectional images, and the availability of many different line widths in a single exposure greatly facilitates this resist characterization process. This mask was designed for testing experimental resists for EUVL, the leading candidate mentioned in the 2007 International Technology Roadmap for Semiconductors (ITRS) as one of the next-generation lithography technologies for the 32 nm node.¹⁸ One of the challenges is the development of dedicated EUV resists, which are expected to allow the printing of 22 nm and smaller node structures, and simultaneously meet the requirements for line-edge roughness and dose. The SEM image of a pattern exposed using this mask in a EUV resist can be seen in Fig. 3(b). The EUV-IL system at PSI has been used extensively for screening these resists.^{9,19,20}

High-resolution electron beam resists, such as PMMA, calixerene, and hydrogen silsesquioxane (HSQ) have been



Fig. 3 The layout of a 20 to 50 nm half-pitch line mask and scanning electron microscopy (SEM) image of its pattern exposed into EUV resist. The interference areas are large and homogenous, suitable for producing cross sections for SEM characterization.



Fig. 4 Line and dot patterns made by EUV-IL exposure in HSQ using different masks: (a) 125 nm half-pitch lines, (b) 175 nm half-pitch dots, (c) and (d) 50 nm half-pitch lines and dots, (e) 11 nm half-pitch lines, and (f) 19 nm half-pitch dots.

the resists of choice in EUV-IL applications.^{10,21,22} There are rapid advances made in the development of chemically amplified EUV resists,⁹ and we expect that these resists will also find use in EUV-IL, further enhancing the throughput of this emerging technique.

Examples of line and dot patterns with various periods that were realized using the EUV-IL tool at PSI are demonstrated in Fig. 4. All of the structures in this figure were exposed in negative tone HSQ resist with various thicknesses. Note the excellent pattern definition and the high quality in terms of line edge roughness. The lines in Fig. 4(e) have a half-pitch of 11 nm, which represents the highest resolution pattern reported with a photolithography method and proves the feasibility of obtaining such small patterns with EUV exposures. The resist thickness for this particular pattern was 20 nm. This result lets us conclude that the blurring effects of photo- or secondary-electrons, if any, are not strong enough to prevent us from reaching 11 nm resolution. Further experiments are underway to explore the ultimate limits of the resolution of EUV exposures.

More advantages of EUV-IL come from the fact that the patterns can be created in a step-and-repeat manner, so that an area of several square millimeters can be exposed in each step. This is considerably faster than conventional focused beam methods, such as EBL or focused ion beam (FIB) lithography. This rapid printing capability allows the production of a large number of samples. Additionally, the mask-based method provides excellent reproducibility, in contrast to serial writing approaches where uniformity problems can be encountered.



Fig. 5 (a) 100 nm period nanowires consisting of 20 nm gold colloids formed on Nb_2O_5 substrate, and (b) 25 nm-wide gold nanowires made using a double stack PMMA/HSQ resist process with shadow metal deposition.

3 Applications

Having reviewed the basic properties of EUV-IL as a nanolithography method, in this section we present several examples of its use in various research areas. The work was performed using the capabilities of the EUV-IL tool at the PSI over a period of about five years. They demonstrate the versatility of this new method and the use of its distinguishing feature, namely the combination of high resolution and high throughput. We do not report on all of the applications of the EUV-IL method that have been demonstrated using the PSI EUV-IL system, which has also led to the produc-tion of nanophotonic devices,^{23,24} nanoimprint stamps,²⁵ holographic fabrication of Fresnel zone plates,^{26,27} and guided self-assembly of block copolymers²⁸ and colloidal particles.²⁹ Thanks to its success, EUV-IL technology has entered the commercial phase through a spin-off company of PSI (see http://www.eulitha.com) that makes specialized samples such as commercially available nanoimprint stamps.

4 Arrays of Metal Nanowires

At the nanometer scale, physical properties of metallic nanowires are expected to deviate significantly from the bulk metal due to confinement and surface effects. For example, the conductivity of metal nanowires changes considerably due to the drastic increase of the surface-to-volume ratio.³⁰ Their mechanical properties, such as the yield strength, also change as their diameter is reduced to some tens of nanometers. The novel properties of nanowires make them candidates for use in nanoelectromechanical systems, for example for electronic circuits, data storage, or biomedical applications.^{31,32}

EUV-IL has been used to produce large arrays of planar metal nanowires. To study the nanowire properties, one needs to produce a substantial number of nanowires and arrange them on a surface in a controlled way. Besides the outstanding resolution, the EUV-IL method is a suitable patterning technique to produce the required large arrays of periodic and homogenous lines in a wide size range for research on these systems.

The goal of the fabrication effort is to produce nanowire arrays with wire widths in the range from sub-10 nm down to 100 nm, over areas of the order of several millimetersquare to study their mechanical, electronic, and optical properties. For the fabrication of metallic nanowires, two different approaches were tried: selective assembly of metal colloids on prepatterned substrates and metallization using a modified lift-off process.

In the first method, EUV-IL-created patterns in resists were used as a template to guide the assembly of Au colloids to form nanowires.³³ The wires were produced on Nb₂O₅-coated surfaces via PLL-g-PEG chemistry. Singlestrand DNA was attached to biotin molecules arranged on 100 nm period lines. Au colloids of diameters 5, 20, or 100 nm, tagged with the complementary DNA strand, were then selectively attached to the surface. Figure 5(a) shows an SEM image of gold wires composed of 20 nm colloids, and demonstrates the selectivity of this chemical recognition process, i.e., Au colloids are only found along the molecularly patterned lines. The lines can be used as the basis for sensitive biosensors for label-free electrical and optical detection based on surface plasmons.

The second metallization method uses a modified lift-off process.³⁴ A double resist layer of PMMA/HSQ was exposed with EUV-IL. A line pattern was formed in the top HSQ layer after development. This pattern was then etched into the underlying PMMA layer. In the subsequent step, a metal film was deposited at an oblique angle on top of the resist lines from both sides. The angle of deposition was adjusted so that no metal was deposited in the spaces between the resist lines. Oblique angle of this shadow evaporation also changed the line width and allowed us to control the width of the final metal nanowire. After lift-off in acetone, gold and chromium nanowires with 10 to 70 nm width and 50 to 100 nm period were obtained. As an example, 20 nm gold wires are shown in Fig. 5(b). At the sub-10 nm scale, the nanowire integrity starts to be limited by the material properties, such as roughness of resist lines and the grain size of the deposited metal. The electrical properties of both the solid metal nanowires and colloidal nanowires are currently being explored.

Gold nanowires placed on a flexible substrate and subjected to a tensile load provided a model system to study their mechanical properties. Large area and uniform nanowire arrays are needed for this method to provide statistically significant data. Using both the metallization and colloid assembly methods described before, we produced arrays of 40 to 60 nm-wide and 20 to 30 nm-thick nano-



Fig. 6 SFM image of a SiGe QD array with an x/y period of 90 nm/100 nm, respectively.

wires on 125 μ m-thick polyimide substrates. A synchrotron-based x-ray diffraction technique was used to measure their (thermo-) mechanical properties. These experiments showed that gold nanowires are significantly stronger than bulk gold but less strong than the thin film of the same thickness. In contrast to bulk gold, strong temperature-dependent mechanical properties are observed for the nanolines. *In-situ* SEM investigations revealed an interesting cracking behavior. Line cracks in Au nanolines on homogenous Cr interlayer thin film were observed at 2.5% strain, but the same nanolines on bare polyimide substrate crack statistically at 6.5% strain.^{35,36}

5 SiGe Quantum Dots

One major drawback of self-assembled quantum dots (QD) lies in their random nucleation, which results in an inhomogeneous distribution of the dot size and position. However, for potential applications like the fabrication of QDbased nanodevices³⁷ or for the demonstration of spintronics and manipulation of qubits,^{38,39} the exact positioning of identical QDs is required. An elegant approach to circumvent the problem of random nucleation is templated selfassembly of semiconductor nanostructures, which combines top-down lithography for definition of the nucleation sites with subsequent epitaxial deposition. Typically, electron beam or optical interference lithography has been used for fabricating prepatterned substrates.^{40,41} However, these methods are limited either in throughput, pattern control, or resolution. In this work EUV-IL was combined with molecular beam epitaxy (MBE) to create Si/Ge quantum dot arrays of high resolution with a remarkably narrow size and height distribution.

The process starts with the EUV-IL exposure of PMMAcoated Si(100) substrates. With EUV-IL, hole patterns with periods ranging from 280 down to 35 nm were generated. Subsequently, the PMMA is developed and the pattern is transferred into the Si substrate by means of reactive ion etching. The SiGe deposition is performed in a solid source MBE system. The atomic force microscopy (AFM) image in Fig. 6 shows the pattern after MBE deposition of a Si buffer layer and a Ge dot layer. The SiGe islands are solely formed in the holes, which results in a highly uniform 2-D array of SiGe QDs. With this technique, perfectly aligned



Fig. 7 SiGe QD crystal with a period of x=90 nm, y=100 nm, and z=11.5 nm. It consists of 11 Ge dot layers separated by 10 Si spacer layers.

SiGe quantum dot arrays with different periods, different QD shapes, and different symmetries have been fabricated.^{42,43} For optical studies, the SiGe quantum dot arrays were investigated by photoluminescence (PL) measurements. Due to the uniformity of the dots, a more efficient PL with much narrower line widths is observed compared to dots grown on unpatterned substrates,⁴⁴ which is of high interest for the realization of Si based optoelectronic devices.

Furthermore, the formation of a 3-D QD crystal is realized by subsequently depositing a stack of Ge layers sepa-⁴⁵ The rated by Si spacer layers onto the 2-D SiGe QD array.⁴³ Si spacer causes a flattening of the surface for the subsequent SiGe QD growth. During the growth of the Ge layers, the QDs nucleate on top of the buried SiGe QDs-a process induced by the local strain field created by the underlying QDs.⁴⁶ Figure 7 shows a SiGe QD crystal with a lateral period of 90 nm in the x direction and 100 nm in the y direction. The vertical period is 11.4 nm. The image is composed of one AFM and two transmission electron microscopy (TEM) images. Perfect vertical ordering of the SiGe QDs is observed. Moreover, a similarly perfect lateral ordering of the SiGe QDs in the top-most layer is seen in the SFM image of Fig. 7. The islands at the top are (105) faceted and have a square base. X-ray diffraction measurements indicate a high structural quality of the SiGe QD crystals.⁴⁵ Recently, QD crystals with lateral periods down to 35 nm have been obtained (unpublished results). It is expected that electronic coupling between adjacent QDs may already occur in such densely packed QD structures. Thus, not only from a structural point of view but also with respect to the band structure, these 3-D ordered QD structures can be regarded as artificial crystals. The EUV-IL technique has been critical to achieve the level of perfection obtained in these samples due to the throughput that allowed the necessary trials on a large number of samples.

6 Arrays of Magnetic Nanoislands

The data storage medium in a hard disk drive is currently a granular magnetic thin film where the grain diameters are typically less than 10 nm. Data are written into the medium by a strong, highly confined magnetic field produced by a



Fig. 8 Schematic diagram of the fabrication of nanoscale magnetic island arrays with EUV-IL and corresponding SEM images of SiO_x pillars with periods of 50 nm (a) before and (b) after Co/Pd multilayer film deposition.⁴⁸

recording head that sets the direction of the magnetization of ~ 100 grains to form a bit. To keep pace with the demand for increased data storage density, the magnetic bit size must be decreased with a commensurate decrease in grain size since, to maintain an adequate signal-to-noise ratio, the number of grains per bit is held approximately constant. However, as the individual grain volumes shrink, thermal activation can lead to unintended reversal of magnetization, since the thermal energy becomes competitive with the magnetic anisotropy energy of the grain. This is the superparamagnetic effect and it defines the grain volume for a given magnetic material, below which the recorded data can be lost due to thermal fluctuations. The magnetic storage industry is therefore seeking ways to delay the onset of superparamagnetism, and one possible route is to record the information in bit patterned media (BPM). A BPM consists of arrays of nanoscale magnetic islands, where each bit of information is stored in an individual island, thus the entire bit volume defines the magnetic anisotropy energy rather than the volume of individual grains. A number of methods have been employed to fabricate suitable nanomagnetic arrays, including electron beam lithography, focused ion beam lithography, and selfassembly methods.⁴⁷ However, EUV-IL has particular advantages for the fabrication of BPM due to the possibility of creating high resolution structures over large areas with high spatial precision and high throughput.

EUV-IL is a relatively new technique for creating patterned magnetic thin films and, due to the short wavelengths of EUV, opens up the possibility of creating arrays of nanostructures with periods corresponding to areal densities greater than 1 Tbit/in², which is beyond the range of optical interference lithography. Initially, EUV-IL combined with electrodeposition was used to create arrays of nickel islands,⁴⁸ and more recently we have developed a method to create 50 nm period arrays of islands of Co/Pd magnetic multilayer films, which have a perpendicular magnetic anisotropy suitable for bit patterned media.⁴⁹ This latter fabrication process is schematically shown in Fig. 8. In a first step, a 30 nm-thick HSQ resist is spin-coated on a silicon substrate. Following EUV-IL exposure through a

grating mask comprised of four linear gratings, the HSO resist is then developed in a tetramethyl ammonium hydroxide solution to produce an array of 30 nm-high SiO_r pillars [Fig. 8(a)]. The magnetic film is then deposited on the prepatterned substrate by dc magnetron sputtering to create arrays of magnetic islands on top of the SiO_x pillars [Fig. 8(b)]. The pillars have a negative sidewall profile due to the strong absorption of EUV light by the HSQ film, so that the top receives a higher exposure dose than the bottom. This negative profile results in shadowing during film deposition, so that there is no magnetic material on the pillar sidewalls. This means that the magnetic islands are magnetically decoupled from the magnetic material in the trenches between the pillars, which is important if they are to be used for magnetic data storage. The growth of the magnetic film on the SiO_x pillars results in a mushroomlike structure (see schematic in Fig. 8). This lateral growth means that typically the size of the magnetic islands is larger than the original SiO_x pillars by a few nanometers, as seen in the SEM images of Fig. 8. The size of the magnetic islands can be tuned by changing the magnetic film thickness and to some extent, for a given mask, by changing the exposure dose values that control the original size of the SiO_x pillars. The 50 nm period magnetic islands shown in Fig. 8 are created over an area of $20 \times 20 \ \mu m^2$ with a very narrow size distribution where the Gaussian sigma is less than 5%, as analyzed using both atomic force microscopy (AFM) images and the SEM images.

Future work will focus on fabricating even smaller magnetic islands. Very recently it has been shown that dot arrays in calixarene resist with periods of 25 nm are achievable with EUV-IL.²¹ This dot period is equivalent to a data storage density of 1 Tbit/in², a key milestone in the data storage roadmap.

7 Chemical Patterning of Self-Assembled Monolayers: Applications in Nanobiotechnology

Self-assembled monolayers (SAMs) are highly ordered organic films with a thickness of one molecule formed on surfaces by chemical adsorption and intermolecular forces.⁵⁰ By using molecules with appropriate functional groups, SAMs form on almost all solid surfaces, providing a versatile route to their functionalization. Usually the formation of a SAM leads to a uniformly covered surface. However, molecular micro-/nanopatterns with specific chemical composition and shape are highly desirable for various research fields and applications, which include phase transitions in wetting layers⁵¹ and polymer blends,¹ hydrodynamic instabilities,⁵² micro-/nanofluidics devices,⁵³ immobilization of nanoparticles,⁵⁴ molecules,⁵⁵ and fabrication of biological chips.⁵⁶ Therefore, many lithographic techniques have been developed for SAMs patterning, such as microcontact printing,⁵⁷ photon-50 or electron irradiation,³³ and scanning-probe-based techniques.³ Lithographic techniques are often employed to induce damage of the exposed regions, which are then exchanged with another SAM of the desired functionality.^{17,59,60} In some cases, however, a controlled modification of SAMs^{55,59,61} is possible, which can directly be used to create chemical patterns. We have shown that EUV-IL^{62,63} presents an example



Fig. 9 (a) Representation of EUV-IL induced chemical patterning of nitro-1, 1'-biphenyl-4-thiol self-assembled monolayer on gold. EUV irradiation induces both the conversion of the terminal nitro groups into amino groups and the lateral cross-linking of the monolayer. (b) AFM micrograph (tapping mode, phase image) of the EUV-IL induced 100 nm periodic nitro/amino line pattern (light/dark lines). (c) Representation of the protein chip assembly on the prepatterned nitro/amino template. Multivalent chelators (dark blue squares) are grown on the amino-terminated areas and loaded with Ni(II) (light blue circles) to immobilize His-tagged (red circles) 20S proteasome complex (11×15 nm). Pristine areas of the NBPT SAM are exchanged with a protein repellent polyethyleneglycol SAM. (d) Topographic AFM image of oriented and specifically immobilized proteasome complexes (period 100 nm); a rectangular area of proteins has been removed from the chip surface in AFM lithography mode (tapping mode, z-scale corresponds to 35 nm), (e) Line profile across the area with and without proteins. For more detail, see Refs. 63 and 65. (Reprinted with permission from Refs. 63 and 65 copyright Wiley-VCH Verlag GmbH and Company). (Color online only.)

of such techniques. We have also demonstrated the utilization of the generated chemical patterns in nanobiotechnology.

EUV irradiation of a 4'-nitro-1,1'-biphenyl-4-thiol (NBPT) SAM on gold induces both the conversion of the terminal nitro groups of NBPT into amino groups and the lateral cross-linking of the underlying aromatic cores, resulting in the formation of an extended supramolecular network⁶³ (Fig. 9). This transformation was studied by x-ray photoelectron spectroscopy, and the secondary electrons, generated in the substrate by EUV light, were identified to be the most likely cause of the observed changes. The complete transformation of nitro into amino groups and the lateral cross-linking occurs at typical radiation doses of $\sim 20 \text{ J/cm}^2$ for the NBPT SAM on gold substrates. Large-area (~4 mm²) chemical line and dot patterns with periods from 2 μ m down to 60 nm consisting of the amino-terminated cross-linked areas and pristine NBPT areas have been generated using EUV-IL. An example of a 100 nm period nitro/amino line pattern is shown in Fig. 9(b). Considering the ultimate resolution of EUV-IL, low

proximity effects, and ~ 1 nm thickness of the resist (NBPT SAM), we believe that the resolution of the chemical patterns can be further improved, and is at present limited by the roughness of the polycrystalline AuPd substrates used and the efficiency of the current exposure masks.

Highly parallel nitro/amino patterns on gold substrates present a versatile tool for the generation of functionally and chemically diverse molecular micro-/nanopatterns. To this end, the terminal amino-groups of the cross-linked areas can be further functionalized, whereas pristine areas of the NBPT SAM can be exchanged with a new SAM of desired functionality.⁶³ This is of particular interest in medical diagnostics and nanobiotechnology for the generation of so-called protein chips,⁵⁶ i.e., solid substrates with specific molecular patterns used for functionally and laterally defined immobilization of proteins and protein complexes. Protein chips facilitate studies of protein functions and interactions, which are of key importance for research in proteomics.⁶⁴ Figure 9(c) schematically presents such a protein chip assembly.⁶⁵ Here, multivalent chelators [cumulated N-nitrilotriacetic acid (NTA) moieties⁶⁶] were chemically grafted to EUV-IL prepatterned cross-linked amino terminated areas. Since the cross-linked areas show an exceptional stability in solution, the areas of the pristine NBPT SAM can be exchanged with a protein-repellent polyethyleneglycol SAM to prevent unspecific adsorption of proteins.⁶³ Loading metal ions [e.g., Ni(II)] to the multivalent chelators results in laterally defined, specific, and reversible binding sides for His-tagged proteins. Figure 9(d) shows an AFM image of a 100 nm periodic line pattern of immobilized proteins,⁶⁵ which was fabricated on the EUV-IL prepatterned NBPT SAM template presented in Fig. 9(b). As a biological model system, the His-tagged barrel-shaped 20S proteasome complex (with lateral dimensions of 11×15 nm) from the archaea Thermoplasma acidophilum⁶⁷ was employed. The His-tags were engineered to give an end-on orientation of proteins [Fig. 9(c)]. Specific and oriented immobilization of the His-tagged proteosome was demonstrated by mechanical removal of proteins from the chip surface with an AFM cantilever, and by complete regeneration of the chip surface.⁶⁵ The line profile over the region with proteasome and without proteasome reveals the height difference [Fig. 9(e)], which is in good qualitative agreement with its specific upright orientation.

The presented examples show the high potential of EUV-IL for chemical patterning of SAMs, which strongly facilitates the generation of diverse large area, highly parallel, and complex molecular micro-/nanopatterns. We expect that the lateral resolution of such patterns will further be improved so that they are capable of immobilization of single biological molecules. This will be a critical development for a manifold of research in molecular biology and nanobiotechnology, including single molecular studies in proteomics and cell adhesion phenomena.

8 Radiation Grafting of Polymer Nanostructures

In polymer technology, radiation grafting is a well known method to tune the properties of polymeric materials, in particular fluoropolymers.⁶⁸ The polymer is exposed to particle or photon radiation with sufficient energy to break bonds in the backbone or side-chains of the polymer. The



Fig. 10 Scheme for the polymer nanografting process: a fluoropolymer foil is first exposed to EUV radiation in the EUV-IL system, leading to a pattern of radicals at the surface of the foil. These radicals serve as initiators for the subsequent graft polymerization of a monomer, resulting in patterned polymer brushes at the surface.

radicals created in this step are then used as initiators for a graft polymerization reaction, in which polymer chains are formed step by step from the corresponding monomer. The graft reaction is carried out either during or after the exposure step. Depending on the parameters used, such as radiation energy, type and concentration of the monomer, type of solvent and reaction time, and temperature, radiation grafting can be used to modify bulk or surface properties of polymers.

EUV radiation is perfectly suited to create radicals for surface grafting of polymers. The photon energy is high enough to create a cascade of secondary electrons with sufficient energy to break chemical bonds, while the process is restricted to a few 10th nanometers near the surface due to the penetration length of EUV photons. The process used to create surface nanostructures with EUV-IL-based polymer grafting is schematically shown in Fig. 10. Typically, 100 μ m-thick films of fluoropolymers such as [poly(ethylene-co-tetrafluoroethylene)] (ETFE), [poly (tetrafluoroethylene-co-hexafluoroproplyene)] (FEP), or [poly(vinylidene fluoride)] (PVDF) are exposed with EUV-IL. The exposure dose determines the density of formed radicals, which in turn determines the density of polymer chains grown in the subsequent graft-polymerization step.

The EUV-radiation grafting differs in many aspects from other lithographic processes, which makes it attractive for a number of applications:

- It is an additive process, i.e., no sacrificial layers such as photoresists or intermediate metal masks are used.
- It can be carried out on polymer surfaces without the need for conducting layers.
- Structures made entirely from polymers are feasible in which the chemical properties of the grafted polymer can be very different from the base material.

Due to its intrinsic chemical amplification, i.e., long polymer chains are grown from single radicals, the process is very sensitive and needs a comparatively low exposure dose. For a number of monomers such as acrylic acid,⁶⁹ styrene,⁷⁰ and various methacrylates,⁷¹ a square root dependence between the exposure dose and the height of the grafted structures was found in the low dose range, indicating that the polymer chains are forced to stretch away from the surface and form so-called polymer brushes. In Fig. 11 the influence of the applied dose on the structure height is shown for two simultaneously grafted 284 nm period dot structures. This pronounced dose-height dependence can be employed to grow surface relief structures by spatial modulation of the exposure dose.⁷⁰

The high rate of the polymerization reaction leads to growth of long polymer chains in short reaction times. Therefore, the overlapping of the polymer chains limits the spatial resolution to about 100 nm. Improvements in the resolution down to the 50 nm range are possible by applying "living-radical" polymerization schemes, which limit the chain growth kinetics, or by postprocessing steps such as thermal or solvent annealing.⁷²

The firm anchoring of the grafted polymer chains enables various chemical derivatization reactions of the grafted chains, including biofunctionalization, e.g., by binding specific protein molecules. For example, the formation of biofunctional and thermoresponsive polymer brush structures has been demonstrated (Fig. 12).^{70,71} Furthermore, the investigations of the growth behavior of the structures enables new insight into surface polymerization mechanisms.^{71,72}

9 Conclusions

Since its introduction about a decade ago,¹⁰ EUV-IL has proven its unique potential as an enabling nanolithography tool. The diverse applications that are outlined in this work



Fig. 11 AFM images of 142 nm half-pitch poly-(glycidyl methacrylate) brush structures grafted onto an ETFE substrate after EUV exposure. Increasing the dose from 5 to 20 mJ/cm² increases the height of the grafted dots from (a) 14 nm to (b) 21 nm.



Fig. 12 Biofunctional microstructured polymer brush: fluorescence image of a 2-µm period structure of poly-(glycidyl methacrylate) after reaction with biotin-pentylamine and binding of fluorescent labeled streptavidin.

demonstrate the versatility of the technique and the various ways it can help in diverse fields. For example, the unrivaled regularity and precision of the periodic structures created by EUV-IL is essential in the self-assembly field (e.g., for GeSi quantum dots) and fabrication of patterned magnetic media. The ability to expose nanopatterns on insulating samples is required for grafting on polymer substrates. Throughput and resolution are cross-cutting benefits of the technique, which help especially in fields like biology, where a great number of experiments need to be performed on similar substrates. With the recent demonstration of a resolution of 11 nm half-pitch patterns, we can expect EUV-IL to continue to serve scientific and technological applications for many years to come.

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