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Letter

# Synthesis and crystal structures of Cs<sub>2</sub>MgD<sub>4</sub> and Cs<sub>3</sub>MgD<sub>5</sub>

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#### Abstract

The title compounds were prepared by sintering stoichiometric mixtures of the binary deuterides in an autoclave under 65 bar pressure at 625 K. Neutron powder diffraction reveals a  $\beta$ -K<sub>2</sub>SO<sub>4</sub> type structure for Cs<sub>2</sub>MgD<sub>4</sub> (space group *Pnma*, a=8.5988(4), b=6.6234(3), c=11.5988(5) Å) and a Cs<sub>3</sub>CoCl<sub>5</sub> type structure for Cs<sub>3</sub>MgD<sub>5</sub> (space group *I*4/*mcm*, a=8.4022(2), c=12.6951(4) Å). In both deuterides magnesium is tetrahedrally coordinated by deuterium. The Mg–D bond distances (1.82–1.88 Å in Cs<sub>2</sub>MgD<sub>4</sub> and 1.84 Å in Cs<sub>3</sub>MgD<sub>5</sub>) are longer by 0.02 Å, on the average, than those in the corresponding rubidium analogues. © 2000 Elsevier Science S.A. All rights reserved.

Keywords: Caesium magnesium deuterides; Saline hydrides; Neutron powder diffraction; Crystal structure

# 1. Introduction

Magnesium is a useful constituent for hydrogen storage materials [1,2]. In combination with alkali metals such as rubidium it forms a series of saline metal hydrides such as RbMgH<sub>3</sub> [3], Rb<sub>2</sub>MgH<sub>4</sub> [4], Rb<sub>3</sub>MgH<sub>5</sub> [4] and Rb<sub>4</sub>Mg<sub>3</sub>H<sub>10</sub> [5]. Caesium analogues such as Cs<sub>2</sub>MgH<sub>4</sub> [6] and Cs<sub>3</sub>MgH<sub>5</sub> [7] are known, but their detailed crystal structures have not yet been investigated. In this work a neutron powder diffraction analysis of the deuterides Cs<sub>2</sub>MgD<sub>4</sub> and Cs<sub>3</sub>MgD<sub>5</sub> is presented. It will be shown that the former crystallises with the  $\beta$ -K<sub>2</sub>SO<sub>4</sub> type structure as was correctly assumed [6], whereas the latter adopts the Cs<sub>3</sub>CoCl<sub>5</sub> type structure rather than the Sr<sub>3</sub>SiO<sub>5</sub> type structure proposed previously [7].

# 2. Experimental

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## 2.1. Synthesis

Stoichiometric mixtures of the binary deuterides CsD and  $\alpha$ -MgD<sub>2</sub> were pressed to pellets and sintered in an autoclave at 625 K under a deuterium pressure of 65 bar

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for 5 days. The reaction products obtained were of white colour. They were pyrophoric and extremely sensitive to air and moisture. The binary deuterides were prepared by deuteration of metallic caesium (STREM 99.9%) and magnesium powder (CERAC 99.6, -400 mesh).

#### 2.2. X-ray diffraction

The  $Cs_2MgD_4$  and  $Cs_3MgD_5$  samples were characterized by X-ray powder diffraction (Philips PW820 powder diffractometer, Cu K $\alpha$  radiation, sealed sample holder). The patterns suggested that the former contained a majority phase with orthorhombic  $\beta$ -K<sub>2</sub>SO<sub>4</sub> type structure, and the latter a phase with body-centred tetragonal Cs<sub>3</sub>CoCl<sub>5</sub> type structure. Attempts to obtain single phase samples by modifying the synthesis conditions failed. The  $Cs_2MgD_4$ sample always contained significant amounts of Cs<sub>3</sub>MgD<sub>5</sub> and unreacted binary deuterides as secondary phases, whereas the Cs<sub>3</sub>MgD<sub>5</sub> sample contained unreacted CsD. Both samples also contained MgO inpurities. The refined cell parameters are listed in Table 1 together with those reported for the corresponding hydrides. The refined metal atom positions (not given here) were in agreement with those found in the corresponding Rb analogues [4].

#### 2.3. Neutron diffraction

Data at room temperature were collected on the high-

Table 1 Cell parameters of ternary caesium magnesium hydrides and deuterides

Compound	a (Å)	<i>b</i> (Å)	<i>c</i> (Å)	$V(\text{\AA}^3)$	Reference
Cs <sub>2</sub> MgH <sub>4</sub>	$8.627(2)^{a}$	6.654(2)	11.662(3)	669.4	[6]
Cs <sub>2</sub> MgD <sub>4</sub>	8.5988(4)	6.6234(3)	11.5988(5)	660.59(5)	Present work
Cs <sub>3</sub> MgH <sub>5</sub>	8.424(2)	_	12.728(3)	903.2	[7]
Cs <sub>3</sub> MgD <sub>5</sub>	8.4022(2)	_	12.6951(4)	896.25(5)	Present work

<sup>a</sup> e.s.d. values in parentheses.

resolution powder diffractometer HRPT [8] at the Swiss spallation neutron source SINQ at PSI in Villigen ( $\lambda =$ 1.886 Å,  $2\theta_{max} = 158.95^{\circ}$ , step size = 0.05° in  $2\theta$ , T = 293K). The samples Cs<sub>2</sub>MgD<sub>4</sub> (1.6 g) and Cs<sub>3</sub>MgD<sub>5</sub> (2.1 g) were enclosed in cylindrical vanadium containers of 8 and 9 mm inner diameter, respectively. Rietveld refinements (program FULLPROF [9]) were performed by taking as starting parameters the X-ray metal position values, and by letting the deuterium positions vary about those of the rubidium analogues. For the Cs<sub>2</sub>MgD<sub>4</sub> sample five additional phases were included in the refinement (Cs<sub>3</sub>MgD<sub>5</sub>,



Fig. 1. Observed (top), difference (middle), and calculated (bottom) neutron diffraction patterns of the  $Cs_2MgD_4$  sample (T=293 K,  $\lambda=1.886$  Å).



Fig. 2. Observed (top), difference (middle) and calculated (bottom) neutron diffraction patterns of the  $Cs_3MgD_5$  sample (T=293 K,  $\lambda=1.886$  Å).

Table 2						
Refinement results <sup>a</sup>	on	neutron	diffraction	data	for	$Cs_2MgD_4$

Site	x/a	y/b	z/c	$B_{\rm iso}({\rm \AA}^2)$
4c	0.1594(6)	1/4	0.4109(6)	2.9(1)
4c	0.4856(6)	1/4	0.6973(4)	1.6(1)
4c	0.2314(6)	1/4	0.0781(6)	1.8(1)
8d	0.3133(5)	0.0209(7)	0.1516(3)	4.7(1)
4c	0.0206(7)	1/4	0.0944(6)	5.6(2)
4c	0.8070(9)	1/4	0.5686(6)	4.3(2)
	Site 4c 4c 4c 8d 4c 4c 4c	Site x/a   4c 0.1594(6)   4c 0.4856(6)   4c 0.2314(6)   8d 0.3133(5)   4c 0.0206(7)   4c 0.8070(9)	Site $x/a$ $y/b$ $4c$ $0.1594(6)$ $1/4$ $4c$ $0.4856(6)$ $1/4$ $4c$ $0.2314(6)$ $1/4$ $8d$ $0.3133(5)$ $0.0209(7)$ $4c$ $0.0206(7)$ $1/4$ $4c$ $0.8070(9)$ $1/4$	Site $x/a$ $y/b$ $z/c$ 4c0.1594(6)1/40.4109(6)4c0.4856(6)1/40.6973(4)4c0.2314(6)1/40.0781(6)8d0.3133(5)0.0209(7)0.1516(3)4c0.0206(7)1/40.0944(6)4c0.8070(9)1/40.5686(6)

<sup>a</sup> T = 293 K, space group *Pnma* (no. 62), Z = 4; e.s.d. values in parentheses.  $R_{\text{Bragg}} = 7.5\%$ ,  $R_{\text{F}} = 5.3\%$ ,  $R_{\text{p}} = 1.9\%$ ,  $R_{\text{wp}} = 2.5\%$ , S = 2.7.

Table 3 Refinement results  $^{\rm a}$  on neutron diffraction data for  $\rm Cs_3MgD_5$ 

Atom	Site	x/a	y/b	z/c	$B_{\rm iso}$ (Å <sup>2</sup> )
Cs1	8h	0.6828(2)	0.1828(2)	0	2.4(5)
Cs2	4a	0	0	1/4	3.1(8)
Mg	4b	0	1/2	1/4	2.2(7)
DI	16 <i>l</i>	0.1240(1)	0.6240(1)	0.1625(1)	5.0(5)
D2	4c	0	0	0	3.5(7)

<sup>a</sup> T = 293 K, space group *I*4/*mcm* (no. 140), Z = 4; e.s.d. values in parentheses.  $R_{\text{Bragg}} = 6.6\%$ ,  $R_{\text{F}} = 5.7\%$ ,  $R_{\text{p}} = 2.7\%$ ,  $R_{\text{wp}} = 3.4\%$ , S = 2.4.

Table 4

Selected interatomic distances (Å) and angles (°) in Cs<sub>2</sub>MgD<sub>4</sub>

Cs1–D2	$3.106(8)^{a}$	D1–Mg	1.877(6)
2D1	3.327(7)	D3	2.972(7)
2D3	3.333(1)	D2	3.013(7)
2D1	3.418(7)	D1	3.035(7)
D3	3.540(9)	Cs2	3.046(6)
2D1	3.619(7)	Cs2	3.179(6)
Mg	3.682(7)	Cs1	3.327(7)
D2	3.859(9)	Cs1	3.418(7)
		D1	3.588(7)
		Cs1	3.619(7)
Cs2-2D1	3.046(6)	D2-Mg1	1.823(8)
D3	3.120(9)	2D1	3.013(7)
D3	3.141(9)	D3	3.105(9)
2D1	3.179(6)	Cs1	3.106(8)
D2	3.397(9)	Cs2	3.397(9)
2D2	3.520(3)	2Cs2	3.520(3)
Mg	3.830(8)	2D1	3.762(7)
		Cs1	3.859(9)
Mg-D3	1.822(9)	D3-Mg1	1.822(9)
D2	1.823(8)	2D1	2.972(7)
2D1	1.877(6)	D2	3.105(9)
D3-Mg-D2	116.9(5)	Cs2	3.120(9)
D3-Mg-D1	106.9(3)	Cs2	3.141(9)
		2Cs1	3.333(1)
		Cs1	3.540(9)

<sup>a</sup> e.s.d. values in parentheses.

Table 5 Selected interatomic distances (Å) and angles (°) in  $Cs_3MgD_5$ 

Cs1-2D2	$3.076(2)^{a}$	D1–Mg	1.845(1)
2D1	3.086(2)	D1	2.948(1)
4D1	3.339(2)	2D1	3.046(2)
2Mg	3.846(1)	Cs1	3.086(2)
		2Cs1	3.339(2)
		2Cs2	3.507(1)
Cs2-2D2	3.174(1)	D2-4Cs1	3.076(2)
8D1	3.507(1)	2Cs2	3.174(1)
4Mg	4.201(1)	8D1	3.914(1)
Mg-4D1	1.845(1)		
4Cs1	3.846(1)		
D1-Mg1-D1	106.01(7)		
D1-Mg1-D1	111.23(5)		

<sup>a</sup> e.s.d. values in parentheses.

CsD,  $\alpha$ -MgD<sub>2</sub> [10], MgO and V) and for the Cs<sub>3</sub>MgD<sub>5</sub> sample two (CsD and MgO). For the former 47 parameters were allowed to vary: one zero correction, six background, six scale factors, 12 profile parameters (four for the  $Cs_2MgD_4$  phase and eight for the five secondary phases), three lattice parameters, 13 positional parameters, and six isotropic temperature factors. For the latter 28 parameters were allowed to vary: one for the zero correction, six for the background, 15 for the Cs<sub>3</sub>MgD<sub>5</sub> phase (one scale factor, two cell, eight atomic, and four peakshape parameters) and six for the secondary phases (four for CsD and two for MgO). The observed, calculated and difference neutron powder patterns are shown in Figs. 1 and 2. The refinement results are summarised in Tables 2 and 3, and selected interatomic distances are listed in Tables 4 and 5. The coordination polyhedra around the various atom sites are represented in Figs. 3 and 4.

## 3. Results and discussion

 $Cs_2MgD_4$  and  $Cs_3MgD_5$  are salt-like compounds. They are isostructural with their rubidium analogues [4] and the corresponding deuterides based on zinc [11,12]. Both compounds contain tetrahedral  $MgD_4^{2-}$  anions, and  $Cs_3MgD_5$  also contains an additional deuteride species D<sup>-</sup> bonded to Cs only. Due to matrix effects the Mg-D bond distances in  $Cs_2MgD_4$  (1.82–1.88 Å) and  $Cs_3MgD_5$  (1.84 Å) are longer by 0.02 Å, on the average, than the corresponding distances in the rubidium analogues  $Rb_2MgD_4$  (1.79–1.87 Å [4]) and  $Rb_3MgD_5$  (1.82 Å [4]). As expected, they are shorter than those for octahedral coordinated Mg in binary  $\alpha$ -MgD<sub>2</sub> (1.94 Å [10]). The bond  $(Cs_2MgD_4)$ : 107-117°; D-Mg-D angles  $Cs_3MgD_5:106-111^\circ$ ) deviate only little from the tetrahedral angle. Caesium has nine-fold (Cs2) and 11-fold (Cs1) deuterium coordinations in  $Cs_2MgD_4$  (Fig. 3), and eight-fold (Cs1) and ten-fold (Cs2) deuterium coordinations in Cs<sub>3</sub>MgD<sub>5</sub> (Fig. 4). The Cs-D bond distances (Cs<sub>2</sub>MgD<sub>4</sub>: 3.11–3.86 Å (Cs1), 3.05–3.52 Å (Cs2);  $Cs_3MgD_5$ : 3.08–3.34 Å (Cs1), 3.17–3.51 Å (Cs2)) are longer by 0.2 Å, on the average, than those in the rubidium analogues (Rb<sub>2</sub>MgD<sub>4</sub>: 2.77-3.97 Å (Rb1), 2.86-3.48 Å (Rb2); Rb<sub>3</sub>MgD<sub>5</sub>: 2.87–3.22 Å (Rb1), 3.06–3.34 Å (Rb2). All deuterium sites have six-fold metal coordinations. They are surrounded by five Cs and one Mg atoms except for D2 in Rb<sub>3</sub>MgD<sub>5</sub> which is surrounded by six Cs atoms. The calculated hydrogen storage efficiencies are 1.37 wt% (40.01 g  $H_2 1^{-1}$ ) for  $Cs_2MgH_4$  and 1.18 wt%  $(37.06 \text{ g H}_2 \text{ l}^{-1})$  for Cs<sub>3</sub>MgH<sub>5</sub>. No fluoride analogues are known. The only compound known in the Cs-Mg-F system is  $Cs_4Mg_3F_{10}$  [13].

As to the potassium analogues only  $K_2MgH_4$  is known to exist [14]. It crystallizes with the  $K_2NiF_4$  structure type



Fig. 3. Metal and deuterium co-ordinations in  $Cs_2MgD_4$ ; D sites numbered; site symmetries m (Cs1, Cs2, Mg, D2, D3) and 1 (D1).

rather than the  $\beta$ -K<sub>2</sub>SO<sub>4</sub> structure type which means that hydrogen in that compound surrounds magnesium in an octahedral rather than a tetrahedral configuration. Attempts to synthesise a hypothetic hydride 'K<sub>3</sub>MgH<sub>5</sub>' containing tetrahedral MgH<sub>4</sub><sup>2-</sup> anions have sofar failed. On the other hand, the corresponding zincates all exist (for M<sub>2</sub>ZnD<sub>4</sub> and M<sub>3</sub>ZnD<sub>5</sub>, M=K, Rb, Cs; see Refs. [11,12,15,16]). Since zinc has a smaller atomic radius than magnesium in these type of compounds, the size ratio between the tetrahedral hydride anions and the metal cations M may be a critical parameter for structural stability. Further experiments are necessary to confirm this hypothesis.

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Fig. 4. Metal and deuterium co-ordinations in Cs<sub>3</sub>MgD<sub>5</sub>; D sites numbered; site symmetries m.2m (Cs1), 422 (Cs2), -42m (Mg), m (D1) and 4/m (D2).

## References

- [1] R.B. Schwarz, MRS Bull. 24 (1999) 40.
- [2] K. Yvon, Proceedings of the International Symposium on Metal– Hydrogen Systems: Fundamentals and Applications, Les Diablerets, Switzerland, August 25–30, 1996, J. Alloys Comp. 253–254 (1997) 1.
- [3] F. Gingl, T. Vogt, E. Akiba, K. Yvon, J. Alloys Comp. 282 (1999) 125.
- [4] M. Bortz, A. Hewat, K. Yvon, J. Alloys Comp. 268 (1998) 173, and references therein; Corrigendum: J. Alloys Comp. 288 (1999) 326.
- [5] F. Gingl, T. Vogt, E. Akiba, K. Yvon, J. Alloys Comp. 284 (1999) L4, and references therein.
- [6] H.H. Park, M. Pezat, B. Darriet, P. Hagenmuller, Chem. Scripta 28 (1988) 447.
- [7] H.H. Park, M. Pezat, B. Darriet, CR Acad. Sci. (Paris) 307 (2) (1988) 555.

- [8] P. Fischer, G. Frey, M. Koch, M. Koennecke, V. Pomjakushin, J. Schefer, R. Thut, N. Schlumpf, R. Buerge, U. Greuter, S. Bondt, E. Berruyer, Proc. of the ECNS '99. Physica B (2000) in press.
- [9] J. Rodriguez-Carvajal, in: Cong. Int. Union of Crystallography, Toulouse, Satellite Meeting on Powder Diffraction, 1990, p. 127.
- [10] M. Bortz, B. Bertheville, G. Böttger, K. Yvon, J. Alloys Comp. 287 (1999) L4.
- [11] M. Bortz, A. Hewat, K. Yvon, J. Alloys Comp. 248 (1997) L1.
- [12] M. Bortz, A. Hewat, K. Yvon, J. Alloys Comp. 253–254 (1997) 13.[13] R.E. Schmidt, J. Pebler, D. Babel, Eur. J. Solid State Inorg. Chem.
- 29 (1992) 679.
- [14] H.H. Park, M. Pezat, B. Darriet, P.R. Hagenmuller, Chim. Minér. 24 (1987) 525.
- [15] M. Bortz, K. Yvon, P. Fischer, J. Alloys Comp. 216 (1994) 39.
- [16] M. Bortz, K. Yvon, P. Fischer, J. Alloys Comp. 216 (1994) 43.